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# Highly efficient ethanol gas sensor based on hierarchical $SnO_2/Zn_2SnO_4$ porous spheres



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ARTICLE INFO	A B S T R A C T			
Keywords: SnO <sub>2</sub> /Zn <sub>2</sub> SnO <sub>4</sub> Hydrothermal method Porous sphere Gas sensor Ethanol	In this work, hierarchucal porous $SnO_2/Zn_2SnO_4$ nanospheres were succesfully prepared via a facile one-step hydrothermal method with subsequent calcination process. Scanning electron microscopy (SEM), and trans- mission electron microscopy (TEM) were employed in order to investigate the structural and morphological properties of the as-prepared composites. The results showed that the $SnO_2/Zn_2SnO_4$ composites were cpmposed of many porous nanospheres with a uniform diameter of about 500 nm. Moreover, the as-prepared products were used as sensing material for the fabrication of gas sensor. The sensing performance of the sensor was system- atically evaluated, and the sensor exhibited excellent ethanol-sensing property. The optimum operating tem- perature was 250 °C with a reponse of 30.5 toward 100 ppm ethanol. Also, the sensor showed good selectivity, stability and a low detection limit of 0.5 ppm (response 1.4). The good sensing performance of $SnO_2/Zn_2SnO_4$ nanospheres can be attibuted to the porous structure as well as the heterojunction formed between $SnO_2$ and $ZnSn_2O_4$ .			

# 1. Introduction

During the last decades, gas sensors based on semiconductor metal oxides have attracted extensive attention in the application of airquality detection, environmental protection, inflammable gas monitoring, human health, and public safety, etc [1–5]. Semiconductor metal oxides due to their merits of easy fabrication, low cost and energy consumption, small in size and good chemical stability have received much scientific attention and regarded as important promising materials for gas sensors. Up to now, many semiconductor metal oxides such as ZnO [6–8],  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> [9–11], In<sub>2</sub>O<sub>3</sub> [12–14], SnO<sub>2</sub> [15–17], WO<sub>3</sub> [18–20] and NiO [21–23] have been successfully developed and used as gas sensing materials, some achivements have been obtained. However, the design and fabrication of new type sensing material for ever increasing the selectivity, sensitivity and decreasing the detection limit still remains a scientific challenge.

In the last decades, some complex oxides have attracted widely interest with the possibility to optimize physical and chemical properties of gas sensors. Some complex oxides including ZnO/SnO<sub>2</sub> [24–26], ZnO/ $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> [27,28],  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/NiO [29–31], In<sub>2</sub>O<sub>3</sub>/SnO<sub>2</sub> [32,33] and CuO/SnO<sub>2</sub> [34–36], etc. have been already reported as gas sensing

materials. Compared with single oxides sensing materials, complex oxides usually have higher sensitivity, better stabily and lower detection limit. However, novel sensing materials with special structure is still needed to study in order to aquire high performance gas sensor. Spinel oxides with a formula of AB2O4 are very promising complex oxides for gas sensing application [37-41]. Zinc stannate (Zn<sub>2</sub>SnO<sub>4</sub>) is an important ternary oxide with an inverse spinel structure and have a band gap of 3.6 eV [42]. Due to its high chemical sensitivity, low visible absorption and excellent optical electronic properties Zn<sub>2</sub>SnO<sub>4</sub> is a proming functional material in various advanced technologies, such as solar cells [43-45], photocatalyst [46], lithium ion battery [47] and gas sensors [48-50]. It is well kown that the gas sensing characteristics are highly dependent on its morpholoy, composition and structure such as porosity, grain size and surface area [51-53]. To date, many nanostructured Zn<sub>2</sub>SnO<sub>4</sub> with different morphologies and structures including nanowires, nanorods, nanospheres and polyhedrons were reported to detect volatile organic gases (VOCs) [49,50,54-57]. However, as far as we know, controlled synthesis of porous SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub> nanospheres with excellent sensing properties toward ethanol has been rarely reported.

In this work, hierachical porous  $SnO_2/Zn_2SnO_4$  nanospheres were

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Fig. 1. Schematic diagram of the experimental procedure.



Fig. 2. (a and b) Schematic diagram of the sensor device.



Fig. 3. XRD pattern of SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub> porous sphere.

successfully prepared via a one-step hydrothermal process. The morphology, microstructure and crystallization properties were conducted by XRD, SEM and TEM techniques. The as-prepared pruducts were fabricated as gas sensor and the gas sensing properties were systematically studied. Compared with  $SnO_2/Zn_2SnO_4$ ,  $Zn_2SnO_4$  and  $SnO_2$  nanoparticles the as-prepared porous  $SnO_2/Zn_2SnO_4$  nanospheres exhibited excellent gas sensing properties toward ethanol. At the optimum operating temperature of 250 °C, the sensor based on porous  $SnO_2/Zn_2SnO_4$  nanospheres have the highest response of 30.5-100 ppm ethanol. The gas sensing machemism was also discussed in detail.

#### 2. Experimental

# 2.1. Synthesis process

All reagents used in the experiments were of analytical grade and directly used without any further purification.

The SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub> composites were synthesized through a facile hydrothermal method. In a typical procedure: 0.284 g of Na<sub>2</sub>SnO<sub>3</sub>·4H<sub>2</sub>O and 0.02 g of trisodium citrate dihydrate (C<sub>6</sub>H<sub>5</sub>Na<sub>3</sub>O<sub>7</sub>·2H<sub>2</sub>O) were dissoved in 10 mL of deionized water (named as solution A), and then 0.438 g of Zn (CHCOO) 2:2H2O was dissolved in a mixture of 2 mL of deionized water and 5 mL of ammonia hydroxide (named as solution B). After stirred for a while, solution A was mixed with solution B and then stirred for another 15 min at room temperature. Then the mixed solution was transffered into a 50 mL of Teflon-lined autoclave and kept at 160 °C for 8 h. After the completion of reaction, the autoclave was allowed to cool down to room temperature. The whilte precipitates were collected by centrifugation, washed with deionized water and absolute ethanol alternately for several times and dried in air at 80 °C for about 10 h. Finally, the precipitate was annealed at 800 °C for 0.5 h in air atmosphere with a heating rate of  $5 \,^{\circ}\text{C}\,\text{min}^{-1}$ , and the porous  $\text{SnO}_{2/2}$ Zn<sub>2</sub>SnO<sub>4</sub> composites were obtained. The synthesis process was displayed in Fig. 1. The synthesis procedure of SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub>, Zn<sub>2</sub>SnO<sub>4</sub> and SnO<sub>2</sub> nanoparticles were described in the supporting information.

#### 2.2. Characterization

The crystal structure of the as-prepared products was examined by X-ray powder diffraction (XRD) on a Rigaku D/Max-2550 V diffractometer using Cu-K $\alpha$  radiation ( $\lambda = 1.54178$  Å). The morphologies and crystal structures of the as-prepared sample were observed by field



Fig. 4. (a-c) SEM and (d-f) TEM images of the as-prepared SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub> porous spheres.



Fig. 5. (a-c) HRTEM images of the as-prepared SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub> porous spheres. (d-g) TEM images of an individual microsphere and the corresponding elemental mapping images.

emission scanning electron microscopy (FESEM) on a JSM-7500 F (JEOL) microscope operating at an accelerating voltage of 15 kV. Transmission electron microscopy (TEM) and high-resolution transmission electron microscopy (HRTEM) observations were carried out on a JEM-2200FS apparatus (JEOL) operating at 200 kV. Elemental mapping imsges was explored by TEM attachment.

# 2.3. Fabrication and measurement of gas sensor

The fabrication process of the gas sensor devices could be described as follows: first, the as-obtained products were mixed with an appropriate amount of deionized water to form a homogeneous slurry, and then the slurry was coated on the surface of an alumina tube with the help of a small brush to form a thick film. The alumina tube is 4 mm in length, 1.2 mm in external diameter, and 0.8 mm in internal diameter, the tube was attached with a pair of gold electrodes and each electrode

was connected with a pair of Pt wires. After drying in air at room temperature, the device was then calcined at 400 °C for 2 h to enhance the stability of the gas sensors. Afterwards, a Ni-Cr alloy coil was inserted into the alumina tube as a heater to control the operating temperature of the sensor by adjusting the heating current. Then the device was welded on a socket. The schematic diagram of such device was shown in Fig. 2a and b. The gas-sensing performance of the sensor was evaluated under laboratory conditions (30 RH%, 23 °C). The measurement was processed by a static process: a given amount of the tested gas was injected into a closed glass chamber, and the sensor was put into the chamber for the measurement of the sensing performance. The response of the sensor is defined as  $R_a/R_g$ , where  $R_a$  and  $R_g$  are the resistance of the sensor in air and in target gas. The definition of response and recovery times is the time taken by the sensor to achieve 90% of the total resistance change in the case of adsorption and desorption of the tested gases [29].



**Fig. 6.** (a) Response of  $\text{SnO}_2/\text{Zn}_2\text{SnO}_4$  porous spheres,  $\text{SnO}_2/\text{Zn}_2\text{SnO}_4$ ,  $\text{Zn}_2\text{SnO}_4$  and  $\text{SnO}_2$  nanoparticles at different working temperature upon exposure to 100 ppm ethanol, (b) Response of the four sensors to various test gases with a concentration of 100 ppm.

#### 3. Results and discussion

#### 3.1. Structural and morphological characterization

The phase compositions of the as-prepared  $SnO_2/Zn_2SnO_4$  samples were identified by X-ray powder diffraction (XRD). As shown in Fig. 3, most of the diffraction peaks can be assigned to spinel  $Zn_2SnO_4$  and the rest peaks to  $SnO_2$ , which indicates that the as-prepared products was a composite of  $Zn_2SnO_4$  and  $SnO_2$ . Moreover,  $Zn_2SnO_4$  and  $SnO_2$  in the composite were in good agreement with the inverse spinel  $Zn_2SnO_4$  (JCPDS: 74–2184) and tetragonal  $SnO_2$  (JCPDS: 41–1445). And no other peaks could be found in the XRD patterns of  $SnO_2/Zn_2SnO_4$  composite, which indicated that the  $SnO_2/Zn_2SnO_4$  composites were of high purity.

Fig. 4a-c is the typical SEM morphology of the  $SnO_2/Zn_2SnO_4$ composites. The as-prepared composites were of well-dispersed with a sphere-like morphology, and the diameter of the composites was about 500 nm. Moreover, from the high-magnification SEM image of Fig. 4c, the as-obtained  $SnO_2/Zn_2SnO_4$  spheres were composed of many nanoparticles and forming a hierarchical structure. TEM measurement was applied to get further information of the structure. As depict in Fig. 4d–f, it is evident that the TEM images showed spherical morphology which was in good accordance with the SEM observations. The  $SnO_2/Zn_2SnO_4$  spheres were uniformly dispersed with a diameter of about 500 nm. Furthermore, in the high-magnified TEM image of Fig. 4f, there are bright spots existed, indicating a porous micro-structure of the as-prepared  $SnO_2/Zn_2SnO_4$  composites. The SEM images of  $SnO_2/Zn_2SnO_4$ , pure  $SnO_2$  and  $Zn_2SnO_4$  nanoparticles were displayed in Fig. S2.

HRTEM observation was further performed to confirm the morphogical and crystalline structure of the  $SnO_2/Zn_2SnO_4$  spheres. Fig. 5a and b displays the high-resolution TEM (HRTEM) images obtained from the marked white rectangles in Fig. 5c. In Fig. 5a and b, the lattice fringes could be clearly observed and the spacing of adjacent lattice fringes were measured to be 0.260 and 0.236 nm, which were corresponded to (311) and (200) lattice plane of  $SnO_2$  and  $Zn_2SnO_4$ , respectively. In addition, TEM elemental mapping is conducted to confirmm the spatial distribution of Sn, Zn, and O in the spherical structure of Fig. 5d. As shown in Fig. 5e-g, it can be found that Sn, Zn, and O were homogeneously co-existed in the hierachical structure.

#### 3.2. Gas sensing properties

Since operating temperature plays an important role on the semiconductor metal oxide based gas sensors, the respones of the sensors based on  $SnO_2/Zn_2SnO_4$  porous spheres,  $SnO_2/Zn_2SnO_4$ ,  $Zn_2SnO_4$  and  $SnO_2$  nanoparticles to 100 ppm ethanol as a function of operating temperature were measured, as displayed in Fig. 6a. As the operating temperature changed from 150 to 350 °C, the responses first increase and then reach the maximum value, afterwards the responses decreased with futher increasing temperature. The optimal operating temperature



Fig. 7. (a) Response of the four sensors to ethanol with different concentrations at 250 °C. (b and c) Corresponding dynamic response curves of the four sensors to different concentrations of ethanol.

#### Table 1

Comparison of ethanol sensing performance of gas sensors based on other material in previous reports.

Sensing material	Morphology	Ethanol Con. (ppm)	Tem. (°C)	Res. (R <sub>a</sub> /R <sub>g</sub> )	Ref.
$\begin{array}{c} Zn_2SnO_4/SnO_2\\ Zn_2SnO_4\\ Zn_2SnO_4\\ SnO_2\text{-}ZnO\\ SnO_2\\ SnO_2\\ SnO_2\\ SnO_2\\ SnO_2/Zn_2SnO_4\\ \end{array}$	Octahedral-like Flower-like Nanowires Nanostructures Nanorods Hollow sphere Porous spheres	100 100 50 100 50 500 100	200 380 500 400 300 350 250	14 30.8 21.6 16 12.4 23.5 30.5	[48] [57] [58] [59] [60] [61] This work

for the four gas sensors was 250 °C with responses of 30.5 (SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub> porous spheres), 16.6 (SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub> nanoparticle), 11.7 (Zn<sub>2</sub>SnO<sub>4</sub> nanoparticle), and 21.6 (SnO<sub>2</sub> nanoparticle), respectively. Obviously, the sensor based on SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub> porous spheres exhibited the highest gas response toward ethanol, and then SnO<sub>2</sub> nanoparticles. It is worth noting that after grinded the SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub> nanoparticles exhibited only half of the response value as compared to SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub> porous spheres.

Selectivity is an important parameter of gas sensors, the selectivity of the sensor based on  $SnO_2/Zn_2SnO_4$  porous spheres,  $SnO_2/Zn_2SnO_4$ ,  $Zn_2SnO_4$  and  $SnO_2$  nanoparticles were investigated as shown in Fig. 6b. The bar graph of the four sensors toward various kinds of volatile organic gases, including ethanol, acetone, methanol, formaldehyde, toluene and xylene with a concentration of 100 ppm at their optimum operating temperature of 250 °C was listed. Clearly, it can be seen that all of the four sensors displayed the highest gas response toward ethanol, and a relatively lower response to the other tested gases. In addition, it is worth noting that for the sensor based on  $SnO_2/Zn_2SnO_4$  porous spheres, the gas response to ethanol, acetone, methanol, formaldehyde, toluene and xylene was 30.5, 12.1, 14.9, 11.2, 3.8, and 3.9, respectively. The gas response value toward ethanol was 2–8 times higher than to other tested gases, which indicated that the sensor based on porous  $SnO_2/Zn_2SnO_4$  spheres possesses a high response and good selectivity to ethanol.

Fig. 7a-c is the sensing beaviour of the four sensors when orderly exposed to different concentrations of ethanol at 250 °C. Fig. 7a is the linear curve of the four sensors with the increasing of ethanol concentration, obviously, the four sensors showed nearly a linear increasing with ethanol concentration varied from 0.5 to 5 ppm and 10 to 100 ppm. Fig. 7b and c is the corresponding dynamic response and recovery curves with the ethanol concentration increased from 0.5 to 5 ppm and 10 to 100 ppm. It can be seen that the four senors exhibited a stepwise increase, and the highest response can be observed from the sensor based on porous SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub> sphere. The corresponding response values were 1.4, 1.7, 2.2, 2.5, 2.7, and 3.0 (Fig.7b) for the ethanol concentration from 0.5 to 5 ppm, and for the concentration from 10 to 100 ppm the response values were 9.9, 11.6, 15.4, 20.6, 23.2, and 30.5 (Fig. 7c). Moreover, the sensor based on porous SnO<sub>2</sub>/ Zn<sub>2</sub>SnO<sub>4</sub> sphere have a low detection limit of 0.5 ppm, which indicates the high sensing property of the sensor. Furthermore, in order to confirm the good sensing characteristic of porous SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub> spheres based gas sensor, a comparison between porous SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub> spheres and other similar ethanol sensing material in reported literatures was summarized in Table. 1 [49,58-62]. The results indicate that the ethanol response value of sensor based on porous SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub> spheres exhibited a relatively higher response and lower working temperature.

Response and recovery characteristic is another parameter of gas



Fig. 8. (a–f) Response transient of the porous  $SnO_2/Zn_2SnO_4$  spheres to 100 ppm ethanol at different working temperature. (g) Response and recovery times of the sensor based on  $SnO_2/Zn_2SnO_4$  spheres at different operating temperature.



Fig. 9. (a) Six cycles of dynamic response curves of porous  $SnO_2/Zn_2SnO_4$  spheres to 100 ppm ethanol at 250 °C. (b) Resistance in air and responses to 100 ppm ethanol of  $SnO_2/Zn_2SnO_4$  spheres as a function of the test days at 250 °C.



Fig. 10. The schematic illustration of ethanol gas sensing mechanism of porous  $SnO_2/Zn_2SnO_4$  spheres.

sensor. The response and recovery properties of the sensor based on porous  $SnO_2/Zn_2SnO_4$  spheres toward 100 ppm ethanol at different operating temperature were investigated as displayed in Fig. 8a–f. It can be seen that the resistance of the sensor was distinctly decreased with the operating temperature increased from 200 to 325 °C, in addition, the response and recovery time was also varied at different temperature. At the optimum operating temperature of 250 °C, the responsed time was 2 s, and the recovery time was 114 s, respectively.

Fig. 7g is the response and recovery time at different operating temperature, the response and recovery time was greatly decreased with increasing operating temperature, and the sensor exhibited a good response and recovery characteristic at the optimal temperature of 250 °C.

As far as we know the reversibility and long-term stability are also important parameters to evaluate gas sensors. Fig. 9a illustrates 6 cycles of response and recovery curve of porous  $SnO_2/Zn_2SnO_4$  sensor to 100 ppm ethanol at 250 °C. The response and recovery curve could be well repeated with similar shape of dynamic transients, demonstrating a good reversibility of the sensor. To further investigate the long-time stability of porous  $SnO_2/Zn_2SnO_4$  based sensor, the response and the resistance in air of the sensor to 100 ppm ethanol at 250 °C during 20 days was measured. As depicted in Fig. 9 b, there is no obvious fluctuation in the resistance and response values during the test days, illustrating good long-term stability of the sensor.

## 3.3. Gas sensing mechanism

For n-type semiconductor metal oxide based gas sensors, the most widely accepted gas sensing machenism is based on the resistance change in the process of adsorption and desorption of gas molecules and chemical reactions on the surface of sensing materials [53,63,64]. As illustrated in Fig. 10, when the sensor is exposed in ambient air, oxygen molecules will adsorb on the surface of porous  $SnO_2/Zn_2SnO_4$  spheres and ionize to negatively charged surface-adsorbed oxygen species by capturing free electrons from the conducting band of  $SnO_2/Zn_2SnO_4$  composites, as shown in Eqs. (1)—(3):

$$O_{2(gas)} \rightarrow O_{2 (ads)}$$
 (1)

$$O_{2(ads)} + e^{-} \rightarrow O^{2-}_{(ads)}$$
<sup>(2)</sup>

$$O_{(ads)}^{2-} + e^{-} \rightarrow 20^{-} _{(ads)} \tag{3}$$

As a result, a thick electron depletion layer will form on the sueface of  $SnO_2/Zn_2SnO_4$  spheres, and a high potential barrier is formed between the adjacent nanograins, leading to an increase of resistance in the sensing material. When the sensor is exposed to reducing gas such as ethanol at a moderate temperature, the ethanol molecules would react with the surface adsorbed oxygen species and the captured electrons will release back to the conduction band, resulting in an increasing conductivity and a deceasing resisitance of the sensor. The reaction process between surface adsorbed oxygen species and ethanol is described as Eq. (4):

$$C_2H_5OH + 60^- \rightarrow 2CO_2 + 3H_2O + 6e$$
 (4)

The excellent gas sensing properties of porous SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub> spheres are mainly attributed to the porous hierarchical structure and the heterojunctions formed between SnO<sub>2</sub> and Zn<sub>2</sub>SnO<sub>4</sub>. Firstly, compared with SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub>, Zn<sub>2</sub>SnO<sub>4</sub> and SnO<sub>2</sub> nanoparticles, the hierarchical  $SnO_2/Zn_2SnO_4$  spheres with porous microstructure could provide a higher accessible surface (as shown in Fig. S3 and Table. S1), and beneficial to gas diffusion and adsorption. Therefore, more surface active sites are available for the reaction between adsorbed oxygen and tested gases, leading to an increasing utilization of the sensing body. While for SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub> nanoparticles the porous hierarchical architecture was destroyed, the oxygen and target gas could only adsorb on the surface of the material, resulting in a low utilization of the sensing body and a relatively low gas response. Another reason is the heterojunction formed between Zn<sub>2</sub>SnO<sub>4</sub> and SnO<sub>2</sub>. Accoding to the literature, heterojunctions formed between different semiconductor oxides could make a significant contribution to the gas sensing performance [50,65,66]. Due to the different band gap energy and work function of Zn<sub>2</sub>SnO<sub>4</sub> and SnO<sub>2</sub>, electrons will be transferred between the two semiconductor oxides, forming heterojunctions at the SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub> interfaces. As the conduction band edge of Zn<sub>2</sub>SnO<sub>4</sub> locates at higher

potential than  $SnO_2$ , the electrons in the conduction band of  $Zn_2SnO_4$  would migrate to the conduction band of  $SnO_2$  until their Femi level become euqal [67,68]. This process will result in the generation of additional electron depletion in the interface of  $SnO_2/Zn_2SnO_4$  composites, which will play an important role in the sensing reactions, and resulting in an enhanced sensing performance.

# 4. Conclusions

In summary, we have successfully prepared hierarchical porous SnO<sub>2</sub>/ Zn<sub>2</sub>SnO<sub>4</sub> spheres via a one-step hydrothermal method with subsequent calcination treatment. The results indicated that the SnO<sub>2</sub>/ Zn<sub>2</sub>SnO<sub>4</sub> spheres were composed of many SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub> nanoparticles forming a hierarchical porous micro-structure. The SnO<sub>2</sub>/ Zn<sub>2</sub>SnO<sub>4</sub> composites were used as gas sensing material and the gas sensing performances of the as-fabricated gas sensor were systematically investigated. The results indicated that compared with SnO<sub>2</sub>/ Zn<sub>2</sub>SnO<sub>4</sub>, Zn<sub>2</sub>SnO<sub>4</sub> and SnO<sub>2</sub> nanoparticles the sensor based on porous SnO<sub>2</sub>/ Zn<sub>2</sub>SnO<sub>4</sub> composites displayed excellent gas sensing properties toward ethanol, including high gas response, good reversibility and outstanding selectivity. Moreover, the sensor showed a low detection limit of 0.5 ppm with a response value of 1.4. The good sensing performance can be mainly attributed to the unique porous structure and heterojunction formed between SnO2 and Zn2SnO4. This work indicates that porous SnO<sub>2</sub>/Zn<sub>2</sub>SnO<sub>4</sub> composites are very promising sensing material for the application of ethanol gas sensor.

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#### Appendix A. Supplementary data

Supplementary material related to this article can be found, in the online version, at doi:https://doi.org/10.1016/j.snb.2018.11.070.

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