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## Phase investigation on zinc–tin composite crystallites†

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**Zinc–tin composite crystallites (cubes and octahedra) synthesized by conventional methods in earlier literature, were proved to have  $\text{ZnSn(OH)}_6$  structures rather than perovskite structure  $\text{ZnSnO}_3$  as reported. Furthermore, according to the tolerance factor  $t$  introduced by Goldschmidt,  $\text{ZnSnO}_3$  can not exist as a perovskite structure.**

Zinc stannate, as a ternary semiconducting metal oxide, has received much attention in the past decade. Thus, various methods<sup>1–6</sup> have been applied to prepare zinc stannate to obtain different morphologies. According to most investigators' viewpoints, zinc stannate exists as two types of oxides with different Zn/Sn/O ratios and crystallographic structures: the spinel-type cubic  $\text{Zn}_2\text{SnO}_4$  and the perovskite  $\text{ZnSnO}_3$ .<sup>7</sup> Spinel  $\text{Zn}_2\text{SnO}_4$  synthesized by various methods has been proved to have a stable and certain state,<sup>8</sup> nevertheless the data for the perovskite  $\text{ZnSnO}_3$  are contradictory and ambiguous. Recently, so-called perovskite  $\text{ZnSnO}_3$  structures were synthesized by some groups. For example, Baoyou Geng *et al.*<sup>9</sup> presented a process for the synthesis of octahedral, truncated octahedral, and 14-faceted polyhedral  $\text{ZnSnO}_3$  microcrystals at low temperature (85 °C). Yi Zeng *et al.*<sup>10</sup> have prepared cubic  $\text{ZnSnO}_3$  nanocages with hierarchical architectures *via* a hexamethylenetetramine (HMT) assisted hydrothermal process. Masahiro Miyauchi *et al.*<sup>11</sup> synthesized single crystalline  $\text{ZnSnO}_3$  nanoparticles by a simple one-step hydrothermal reaction. Guanxiang Ma *et al.*<sup>12</sup> have reported a pure-phase synthesis of well-defined faceted cubic  $\text{ZnSnO}_3$  microcrystals in a hydrothermal synthesis. It is worth noting that all the so-called  $\text{ZnSnO}_3$  powders had a small peak at  $2\theta = 43.5^\circ$  in their XRD patterns,<sup>9–12</sup> however, in the standard spectrum of the perovskite  $\text{ZnSnO}_3$  structure (JCPDS No. 11-0274), such a small diffraction peak can not be found. On the other hand, this small peak at  $2\theta = 43.5^\circ$  can be found in the standard spectrum of the  $\text{ZnSn(OH)}_6$

structure (JCPDS No. 73-2384), for which most of the main diffraction peaks are the same as those of the perovskite  $\text{ZnSnO}_3$  (JCPDS No. 11-0274) except for this small peak. In this work, a series of experiments (refer to ref. 9 and 12) and analyses were carried out to identify the phase of the as-synthesized zinc–tin composite crystallites. Our results indicate that these composite crystallites are the  $\text{ZnSn(OH)}_6$  structures rather than the perovskite  $\text{ZnSnO}_3$  as reported.

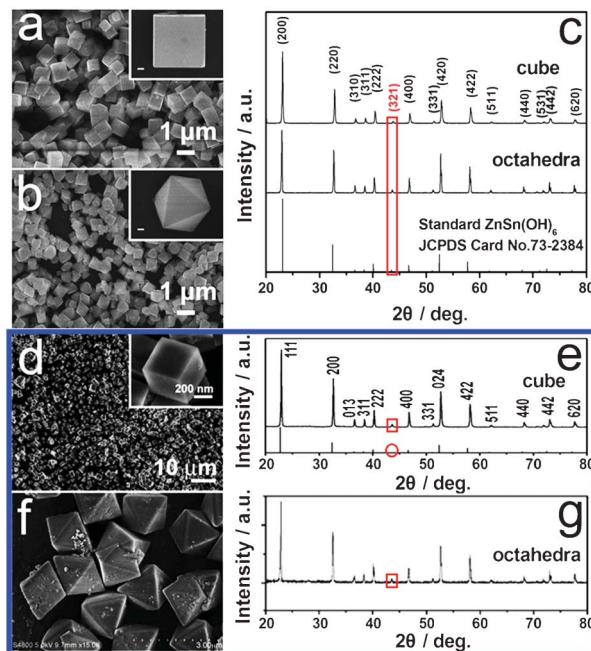
All the reagents in the experiment were of analytical grade and were used without further purification. The cubic and octahedral crystallites were prepared using the same synthetic methods and conditions (raw materials, solution concentration and experimental conditions) as those in ref. 12 and 9, respectively, except for the suppliers of the raw materials (see the ESI† for the experimental details). The X-ray diffraction (XRD) patterns were recorded on a Rigaku D/Max-2500 diffractometer with  $\text{CuK}\alpha$  radiation ( $\lambda = 1.5406 \text{ \AA}$ ). Field emission scanning electron microscopy (FESEM) observations were carried out using a JEOL JSM-7500F microscope with an accelerating voltage of 15 kV. Thermogravimetric (TG) analysis and differential scanning calorimetric (DSC) measurements were carried out with a NETZSCH STA 449F3 simultaneous thermogravimetric analyzer under air in the temperature range 30 °C to 800 °C, with a heating rate of 10 °C min<sup>−1</sup>. The FTIR spectra were recorded on a Nicolet 6700 Fourier transform infrared spectrometer, at wavenumbers 400–4000 cm<sup>−1</sup> using KBr pellets.

Fig. 1(a) and (b) show the SEM images of the as-obtained cubic and octahedral crystallites in our experiment. It can be seen that their morphologies are similar to those presented in Fig. 1(d) and (f), respectively, which were copied from ref. 12 and 9 directly for a comparison. Fig. 1(c) shows the XRD patterns of the as-obtained cubic and octahedral samples. All of the diffraction peaks can be indexed to the cubic  $\text{ZnSn(OH)}_6$  with the calculated lattice parameter  $a = 7.80 \text{ \AA}$ , which is in good agreement with the literature value (JCPDS No. 73-2384), including the small peak (321) at  $2\theta = 43.5^\circ$ , labeled in the figure with a red rectangle. No other crystalline phases or obvious peaks of impurities were detected. Fig. 1(e) and (g) show the XRD patterns of the cubic and octahedral samples prepared by the authors of ref. 12 and 9. They thought that their as-synthesized products were the pure-phase

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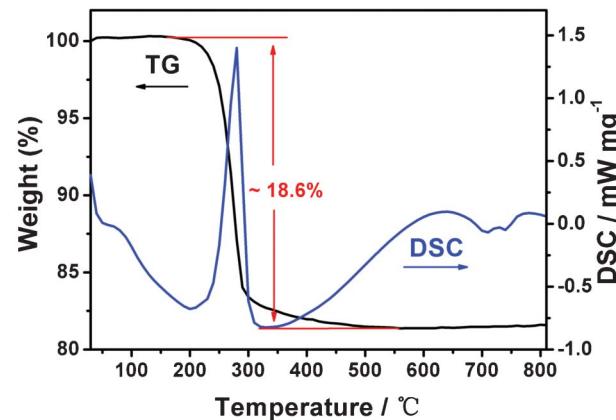
† Electronic supplementary information (ESI) available: Experimental details, TG and DSC curves, phase transition and FTIR data of the octahedral crystallites. See DOI: 10.1039/c3ra41767k



**Fig. 1** SEM images for (a) cubic and (b) octahedral crystallites; XRD patterns (c) of the as-obtained cubic crystallites, octahedral crystallites and the standard XRD pattern of  $\text{ZnSn}(\text{OH})_6$  (JCPDS No. 73-2384). The insets of (a) and (b) show the corresponding high magnification images with a scale bar of 100 nm. SEM image (d) and XRD pattern (e) for the cubic crystallites from ref. 12, SEM image (f) and XRD pattern (g) for the octahedral crystallites from ref. 9, as circumscribed in the blue frame.

perovskite  $\text{ZnSnO}_3$  structures (JCPDS No. 11-0274). However, it is clear that there is a small diffraction peak at  $2\theta = 43.5^\circ$  in the XRD of their products, while such a small peak does not exist in the standard spectrum of the so-called perovskite  $\text{ZnSnO}_3$  structure (JCPDS No. 11-0274) as shown in Fig. 1(e) with a red circle. Contrarily, this is the same as our results; the XRD patterns can be indexed to the standard  $\text{ZnSn}(\text{OH})_6$  structure (JCPDS No. 73-2384), including the small peak at  $2\theta = 43.5^\circ$ . Based on the above analysis, these zinc-tin composite crystallites may be not perovskite  $\text{ZnSnO}_3$ , but  $\text{ZnSn}(\text{OH})_6$  structures.

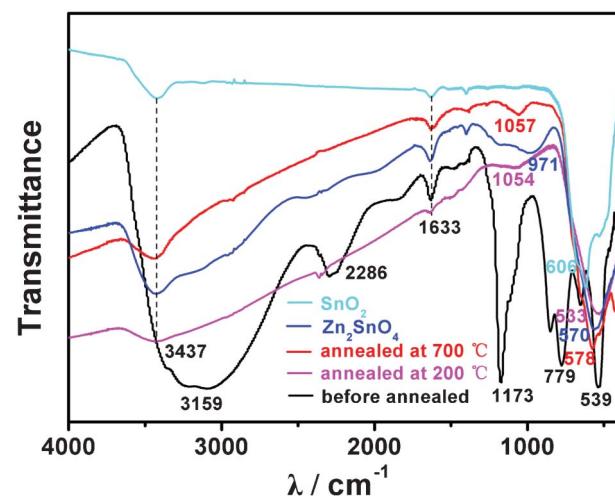
In order to validate our point of view, thermogravimetric (TG) and differential scanning calorimetric (DSC) measurements were carried out under air in the temperature range  $30^\circ\text{C}$  to  $800^\circ\text{C}$ , with a heating rate of  $10^\circ\text{C min}^{-1}$ . The TG and DSC curves of the as-obtained cubic crystallites are shown in Fig. 2. If the samples are perovskite  $\text{ZnSnO}_3$  structures, there should be only a small weight loss, caused by the physisorbed water in the air. However, if the samples are  $\text{ZnSn}(\text{OH})_6$  structures, there should be a dehydration procedure ( $2\text{ZnSn}(\text{OH})_6 \rightarrow \text{Zn}_2\text{SnO}_4 + \text{SnO}_2 + 6\text{H}_2\text{O}$ ) through the heating treatment, with a theoretical proportion of 18.88% for  $3\text{H}_2\text{O}$  in the  $\text{ZnSn}(\text{OH})_6$ . It can be seen that the TG curve presents an obvious weight loss stage, of about 18.6%, in the temperature range  $200^\circ\text{C}$ – $400^\circ\text{C}$ , as shown in Fig. 2. This matches the theoretical weight loss value in the dehydration process of  $\text{ZnSn}(\text{OH})_6$ . This result further proves the fact that a mass of hydroxyl groups exists in the as-obtained samples. These



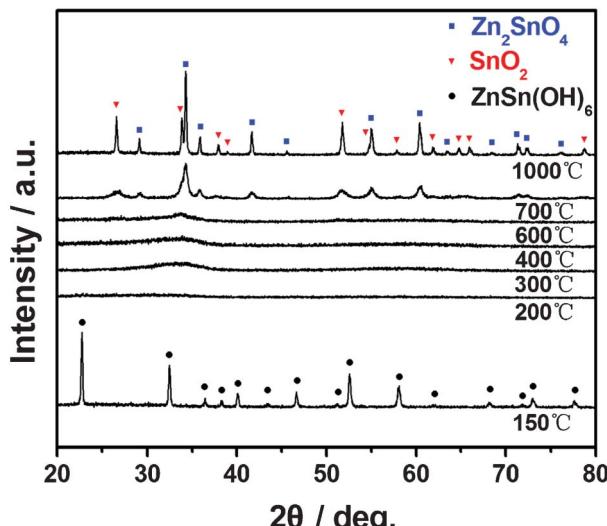
**Fig. 2** TG and DSC curves of the as-obtained cubic crystallites in air.

crystallites are not  $\text{ZnSnO}_3$  but  $\text{ZnSn}(\text{OH})_6$  structures. The DSC curve exhibits an endothermic peak at  $280^\circ\text{C}$ , which can be attributed to the weight loss stage. In addition, there are several exothermic peaks between  $600^\circ\text{C}$  and  $800^\circ\text{C}$ , which can possibly be attributed to the phase transition process. This point will be confirmed by the XRD patterns of the samples annealed at different temperatures (Fig. 4).

FTIR spectra were then recorded in the range  $400$ – $4000\text{ cm}^{-1}$  using KBr pellets, to further verify the existence of hydroxyl groups in the as-obtained samples. Fig. 3 displays the FTIR spectra of vacuum oven dried cubic crystallites before being annealed, annealed at  $200^\circ\text{C}$  and annealed at  $700^\circ\text{C}$ . Furthermore, the FTIR spectra of pure  $\text{Zn}_2\text{SnO}_4$  and  $\text{SnO}_2$  are also presented for comparison. Pure  $\text{Zn}_2\text{SnO}_4$  and  $\text{SnO}_2$  were obtained according to the literature.<sup>12,13</sup> It is specially noted that the samples were pretreated in a vacuum oven at  $100^\circ\text{C}$  for 12 h, just before recording the FTIR spectra, to remove the physisorbed water on the surface of the samples. From Fig. 3, it can obviously be seen that the strong peak at  $3159\text{ cm}^{-1}$  of the cubic crystallites before being



**Fig. 3** FTIR spectra of the vacuum oven dried cubic crystallites before being annealed, annealed at  $200^\circ\text{C}$ , annealed at  $700^\circ\text{C}$ , pure  $\text{Zn}_2\text{SnO}_4$  and  $\text{SnO}_2$ .



**Fig. 4** XRD patterns of the cubic crystallites annealed in air at different temperatures from 150 °C to 1000 °C.

annealed can be attributed to an O-H stretching, which agrees with the literature available on hydroxides, where the hydroxyl ion is among the main components of the structural units of minerals and synthetic hydroxides.<sup>14</sup> The weak peak at 3437 cm<sup>-1</sup> of the other four curves can be attributed to the physisorbed H<sub>2</sub>O molecules on the crystallites surface. This significant difference in the FTIR spectra strongly proves the existence of the hydroxyl groups in the as-synthesized samples again. The peaks observed at wave numbers 2286 cm<sup>-1</sup>, 1173 cm<sup>-1</sup>, 779 cm<sup>-1</sup> and 539 cm<sup>-1</sup> may be attributed to the vibration of the M-OH or M-OH-M groups for the structure of ZnSn(OH)<sub>6</sub>. The peaks at 1057 cm<sup>-1</sup> and 578 cm<sup>-1</sup> may be attributed to the vibration of the M-O or M-O-M groups for SnO<sub>2</sub>-Zn<sub>2</sub>SnO<sub>4</sub>.<sup>15</sup> For pure Zn<sub>2</sub>SnO<sub>4</sub> and SnO<sub>2</sub>, the peaks at 570 cm<sup>-1</sup>, 606 cm<sup>-1</sup> and 971 cm<sup>-1</sup> can be attributed to the vibration of the M-O group. Moreover, the small peaks at 1633 cm<sup>-1</sup> in these five spectra can be attributed to the bending vibration and stretching vibration modes of the hydroxyl group in the physisorbed H<sub>2</sub>O molecules.<sup>16</sup>

To investigate the phase transition process and explore whether we obtained the pure-phase perovskite ZnSnO<sub>3</sub> structure, the as-synthesized samples were annealed in air in a muffle furnace at different temperatures. The XRD patterns of the cubic crystallites annealed in air from 150 °C to 1000 °C are shown in Fig. 4. When the samples were annealed at 150 °C, the diffraction peaks can still be indexed to the standard ZnSn(OH)<sub>6</sub> structure (JCPDS No. 73-2384). However, when the annealing temperature was varied in the range 200–600 °C, the samples became amorphous. When the annealing treatment was carried out at 700 °C, the sample became a complex phase, which can be indexed to the standard XRD patterns of Zn<sub>2</sub>SnO<sub>4</sub> (JCPDS No. 24-1470) and SnO<sub>2</sub> (JCPDS No. 41-1445) without the relative intensity variation and impurity peaks, as marked in Fig. 4 with different signs (see Fig. S2(b) in the ESI† for details). When the annealing temperature was raised to 1000 °C, the diffraction peaks of the complex phase were more obvious. It is noteworthy that the perovskite ZnSnO<sub>3</sub> structure (JCPDS No.

11-0274) did not appear in this whole process. These results further confirm our viewpoint that the as-prepared products were the ZnSn(OH)<sub>6</sub> structures rather than the perovskite ZnSnO<sub>3</sub>. Moreover, Goldschmidt<sup>17</sup> introduced the tolerance factor *t* to describe the characteristic distortion of an ABO<sub>3</sub> perovskite structure,  $t = (R_A + R_O)/\sqrt{2} (R_B + R_O)$ , where R<sub>A</sub>, R<sub>B</sub> and R<sub>O</sub> are the ionic radii of A, B and O, respectively. For ABO<sub>3</sub> to exist as a perovskite structure, *t* must be in the range  $0.77 < t < 1.1$ . The tolerance factor *t* is just about 0.72 for ZnSnO<sub>3</sub>, with the ionic radii of Zn<sup>2+</sup> (0.74 Å), Sn<sup>4+</sup> (0.71 Å) and O<sup>2-</sup> (1.4 Å). From this point of view, ZnSnO<sub>3</sub> can not exist as a perovskite structure since the ionic radius of Zn<sup>2+</sup> is too small.

The TG, DSC (see ESI,† Fig. S1), XRD (see ESI,† Fig. S2) and FTIR (see ESI,† Fig. S3) data of the octahedral crystallites are similar to those of the above-mentioned cubic crystallites, which demonstrates the fact that hydroxyl groups exist in the as-synthesized products. This proves that the zinc-tin composite crystallites are the ZnSn(OH)<sub>6</sub> structures rather than the perovskite ZnSnO<sub>3</sub>.

In summary, zinc-tin composite crystallites (cubic and octahedral crystallites) were prepared by conventional methods, refer to ref. 12 and 9, respectively. From the XRD, TG, DSC and FTIR analyses, it has been proved that these composite crystallites are not the perovskite ZnSnO<sub>3</sub> but ZnSn(OH)<sub>6</sub> structures. This viewpoint is different from the previous reports. Furthermore, these hydroxides, when they were annealed in the range 150 °C–1000 °C, were changed from the ZnSn(OH)<sub>6</sub> structure to an amorphous material at 200 °C and to a SnO<sub>2</sub>-Zn<sub>2</sub>SnO<sub>4</sub> complex phase at 700 °C. It is worth noting that the perovskite ZnSnO<sub>3</sub> structure did not appear in the whole annealing process.

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