



## Enhancement of NO<sub>2</sub> gas sensing response based on ordered mesoporous Fe-doped In<sub>2</sub>O<sub>3</sub>



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### ABSTRACT

Ordered mesoporous Fe-In<sub>2</sub>O<sub>3</sub> has been synthesized via a nanocasting route using the three-dimensional (3D) cubic mesoporous KIT-6 silica as a hard template. Various techniques including X-ray diffraction (XRD), nitrogen adsorption-desorption and transmission electron microscopy (TEM) as well as X-ray photoelectron spectroscopy (XPS) were employed for the material characterization. All the results demonstrate that the obtained Fe-doped In<sub>2</sub>O<sub>3</sub> displays the high surface area, ordered mesoporous structure and well crystallite. The NO<sub>2</sub> sensing properties of the sensors based on mesoporous pure In<sub>2</sub>O<sub>3</sub> and Fe-doped In<sub>2</sub>O<sub>3</sub> samples were detected by a static test system. The sensor utilizing mesoporous Fe-doped In<sub>2</sub>O<sub>3</sub> exhibits much higher response to NO<sub>2</sub> gas compared to that using mesoporous pure In<sub>2</sub>O<sub>3</sub> prepared by the same nanocasting technique. The excellent gas sensing property of mesoporous Fe-doped In<sub>2</sub>O<sub>3</sub> is associated with its higher surface area and pore volume which induce highly effective surface interaction between the target gas molecules and the surface active sites. Moreover, the improved sensing behavior to NO<sub>2</sub> has been linked to the incorporation of iron species on the sensor, favoring the effective adsorption of NO<sub>2</sub> molecules on the surface. All the factors are obviously beneficial for enhancing the gas-sensing performance.

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### 1. Introduction

NO<sub>2</sub>, a typical pollution gas with pungent odor, generating from combustion chemical plants and automotive emissions, plays a major role in the formation of acid rain, photochemical smog and ground-level ozone, all of which are potentially harmful. The detection of nitrogen dioxide [1,2] becomes an exceedingly significant task for monitoring environmental pollution. Various types of techniques [3–7] have been developed to detect NO<sub>2</sub> gas, including semiconductor type, optical fiber, solid electrolyte gas sensor and laser-induced fluorescence instrument. Among them, gas sensors based on metal oxide semiconductor (MOS) [8–12] such as In<sub>2</sub>O<sub>3</sub>, Fe<sub>2</sub>O<sub>3</sub>, SnO<sub>2</sub>, ZnO, and WO<sub>3</sub> are predominant gas detecting devices for domestic, laboratorial and commercial applications, owing to their miniaturization, easy fabrication, low cost, and good long-term stability. The mechanism [13,14] of semiconductor gas sensor is based on surface-chemical interaction between gas molecules and the sensor materials, thus, they are significantly influenced by

the morphology and structure of sensing materials. However, the bulk materials or compact films with low specific surface areas limit the potential applications in gas sensing. Recently, significant research effort has been focused on improving the sensing performance of oxides semiconductor through manipulating their structure morphologies [15–18]. In particular, the ordered mesoporous metal oxides [19–23] synthesized through nanocasting strategy with hard templates have been established as a facile route to obtain gas sensing materials with optimized properties. Owing to their high specific surface area and ordered topological architecture, they can offer more active sites for oxygen adsorption and facilitate high accessibility for the gas molecules to the sensing materials.

Indium oxide, n-type semiconductor material with a wide band gap ( $E_g = 3.55\text{--}3.75\text{ eV}$ ), plays an extraordinary role in a wide domain of useful applications, including gas sensor, catalyst, optoelectronic devices, as well as dye-sensitized solar cells. In<sub>2</sub>O<sub>3</sub> as a promising candidate for gas sensor has been extensively applied in detecting NO<sub>2</sub> [24–27], O<sub>3</sub> [28], CH<sub>4</sub> [29], CO<sub>2</sub> [30] and C<sub>2</sub>H<sub>5</sub>OH [31]. In recent years, different synthesis routes have been developed to improve the sensing properties. Apart from increasing the specific surface area, the gas-sensing characteristics toward a particular

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target gas can also be further improved by surface modification [32–36], such as doping with Pd, Pt, Au, Cu, Fe or their oxides. The additives are chosen on the basis of their ability for enhancing the interaction between the target gas molecules and the sensing material.

Herein, we present an ordered mesoporous Fe-doped  $\text{In}_2\text{O}_3$  prepared via the nanocasting route by using 3D cubic KIT-6 (space group Ia3d) mesoporous silica as a hard template, with indium and ferric nitrate salts as precursors. The morphological and structural properties of the material, as well as the chemical state of the additive are analyzed. The sensing properties of mesoporous pure  $\text{In}_2\text{O}_3$  and Fe-doped  $\text{In}_2\text{O}_3$  samples for  $\text{NO}_2$  were measured. The introduction of metallic Fe is found to be effective to enhance the sensing characteristic of mesoporous  $\text{In}_2\text{O}_3$  sensor toward  $\text{NO}_2$ , because the doping of iron species favors the effective adsorption of  $\text{NO}_2$  on the sensor surface [37,38] and facilitates the reaction between oxygen species and  $\text{NO}_2$  gas.

## 2. Experimental

### 2.1. Preparation of materials

The mesoporous KIT-6 silica with cubic Ia3d symmetry was prepared according to the previously report [39]. In a typical synthesis process, 6 g of Pluronic P123 ( $\text{EO}_{20}\text{PO}_{70}\text{EO}_{20}$ , MW = 5800, Sigma–Aldrich) was dissolved in 217 g of distilled water and 11.2 g of HCl (37%). Then, 6 g of butanol was added under stirring at  $35^\circ\text{C}$  for 1 h. 12.9 g of TEOS was added into the above solution and subsequently the mixture was left under stirring for 24 h at  $35^\circ\text{C}$ . The resulted gel was transferred into a closed Teflon-lined stainless steel autoclave and heated at  $100^\circ\text{C}$  for another 24 h under a static condition. Finally, the solid product obtained after hydrothermal treatment was filtered off, washed with deionized water, dried at  $100^\circ\text{C}$  and sintered at  $550^\circ\text{C}$  for 6 h.

Mesoporous Fe-doped  $\text{In}_2\text{O}_3$  was synthesized via nanocasting method. 1.72 g of  $\text{In}(\text{NO}_3)_3 \cdot 4.5\text{H}_2\text{O}$  and 0.09 g  $\text{Fe}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$  were dissolved in 10 ml of absolute ethanol. After stirring for 30 min, 0.5 g of KIT-6 was added to the above solution and stirring at  $40^\circ\text{C}$  until all the solvent was evaporated, dried at  $80^\circ\text{C}$ , and calcined at  $300^\circ\text{C}$  for 3 h. To achieve higher loadings, the filling procedure was repeated following the same conditions but the amount of precursor was reduced by half. Sequentially, the resultant powder was

calcined at  $500^\circ\text{C}$  for 3 h. Finally, the silica template was removed using excess NaOH solution (2 M). The obtained slurry was centrifuged, copiously washed and dried. Mesoporous pure  $\text{In}_2\text{O}_3$  was synthesized by a similar procedure.

### 2.2. Characterization

X-ray diffraction (XRD) patterns were recorded on a Rigaku D/MAX-2550 diffractometer using  $\text{Cu-K}\alpha$  radiation ( $\lambda = 0.15418 \text{ nm}$ ). Nitrogen adsorption-desorption isotherms were measured using a Micromeritics Gemini VII surface area and porosity system by high purity nitrogen as adsorbate at 77 K. The specific surface area was estimated by the five point Brunauer–Emmett–Teller (BET) method, and the pore size distribution was obtained by using the Barrett–Joyner–Halenda (BJH) analysis. Transmission electron microscopy (TEM) images were performed on a JEOL TEM-3010 instrument, operating at an acceleration voltage of 200 kV. Analysis of the X-ray photoelectron spectroscopy (XPS) was performed on Thermo ESCALAB 250 spectrometer.

### 2.3. Sensor fabrication and measurement

The as-synthesized samples were ground and mixed with deionized water in an agate mortar to form paste, and then the paste was coated onto an alumina tube on which a pair of Au electrodes was previously printed and Pt lead wires attaching to the Au electrodes were used for connecting with the measure instrument. Then, the devices were dried under IR radiation for 1 h, sintered at  $500^\circ\text{C}$  for 2 h, and a Ni–Cr alloy coil was subsequently inserted into the alumina tube as a heater to keep the operating temperature. The sensing properties were carried out using a static test system, as diagrammed in Fig. 1, under laboratory conditions ( $40\% \pm 10\% \text{ RH}$ ,  $25 \pm 1 \text{ c}$ ). The sensor was put into the test chamber, and a given amount of the target gas was injected into the test chamber for the measurement of the sensing performance. The electrical properties of the sensor were recorded by a digital multimeter (Fluke 8846A). The sensor response is defined as  $S = R_g/R_a$  (reducing gas) and  $S = R_g/R_a$  (oxidizing gas), where  $R_a$  is the resistance in air and  $R_g$  is the resistance in the target gas. The response and recovery time were defined as the time taken by the sensor to achieve 90% of the

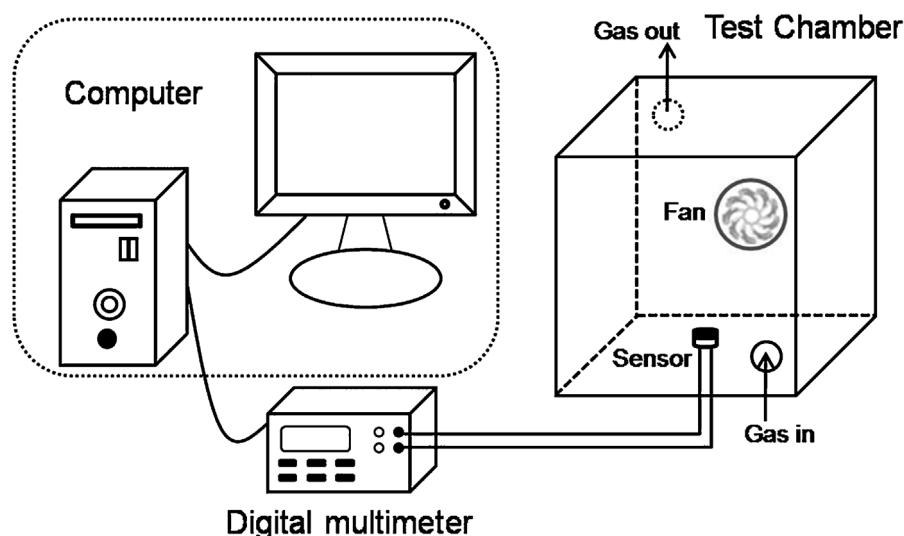


Fig. 1. Schematic illustration of sensor testing system.

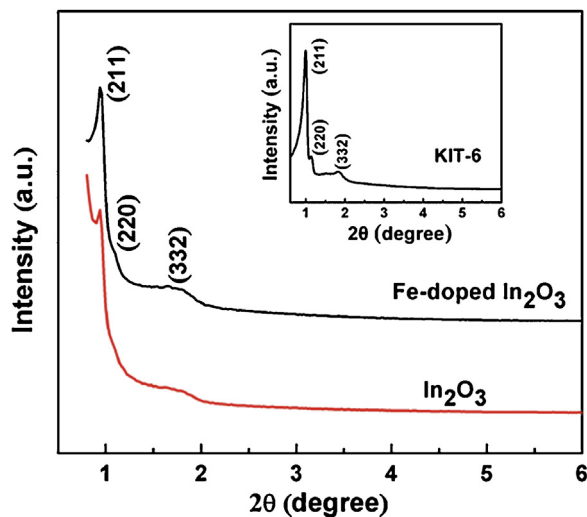


Fig. 2. Low-angle XRD patterns of mesoporous  $\text{In}_2\text{O}_3$  and Fe-doped  $\text{In}_2\text{O}_3$ , as well as the hard template KIT-6 (inset).

total resistance change in the case of adsorption and desorption, respectively.

### 3. Results and discussion

#### 3.1. Structure and morphology of as-prepared materials

Fig. 2 shows the low-angle X-ray diffraction (XRD) patterns of mesoporous  $\text{In}_2\text{O}_3$  and Fe-doped  $\text{In}_2\text{O}_3$  samples, as well as the hard template KIT-6 (inset). The patterns of  $\text{In}_2\text{O}_3$  and Fe-doped

$\text{In}_2\text{O}_3$  display the (2 1 1) (2 2 0) and (3 3 2) reflections, implying the samples are the negative replica of the KIT-6 silica matrix with periodically ordered cubic Ia3d symmetry mesostructure. In the case of KIT-6 matrix, the well-resolved diffraction is sharp, whereas the replicated  $\text{In}_2\text{O}_3$  and Fe-doped  $\text{In}_2\text{O}_3$  show somewhat broader and less well-resolved peaks, which indicate that the long-range order in the  $\text{In}_2\text{O}_3$  and Fe-doped  $\text{In}_2\text{O}_3$  samples are still rather low.

The wide-angle XRD patterns of mesoporous  $\text{In}_2\text{O}_3$  and Fe-doped  $\text{In}_2\text{O}_3$  samples are illustrated in Fig. 3(a). Both the diffraction patterns can be indexed to the body-centered cubic bixbyite-type crystalline phase  $\text{In}_2\text{O}_3$  (JCPDS No. 06-0416), while no peaks were attributed to any iron or iron-based compounds in the patterns of the mesoporous Fe- $\text{In}_2\text{O}_3$  samples. The positions of the corresponding diffraction peaks from Fe-doped  $\text{In}_2\text{O}_3$  sample are slightly shifted to higher angles compared to the pure  $\text{In}_2\text{O}_3$  sample, especially for those corresponding to (2 2 2) plane, as shown in the enlarged partial view of the XRD patterns (Fig. 3(b)). The hardly discernible diffraction of Fe-related can be attributed to the Fe ions embedding into the  $\text{In}_2\text{O}_3$  host structure. The lattice parameters of the mesoporous  $\text{In}_2\text{O}_3$  and Fe- $\text{In}_2\text{O}_3$  were calculated to be 1.010 and 1.006 nm, respectively, which is consistent with the ionic radius of  $\text{Fe}^{3+}$  is smaller than that of  $\text{In}^{3+}$ . The crystallite size of samples can be calculated by the Scherrer equation [40],

$$D = \frac{0.9\lambda}{\beta \cos \theta} \quad (1)$$

where  $D$  is the crystallite size,  $\lambda$  is the wavelength of X-ray radiation,  $\beta$  is the full width at half-maximum intensity (fwhm in radians),  $\theta$  is diffraction angle.

The average crystallite size of the mesoporous  $\text{In}_2\text{O}_3$  and Fe-doped  $\text{In}_2\text{O}_3$  are approximately 16.8 and 14.0 nm, respectively. All the well-resolved characteristic patterns indicate that well

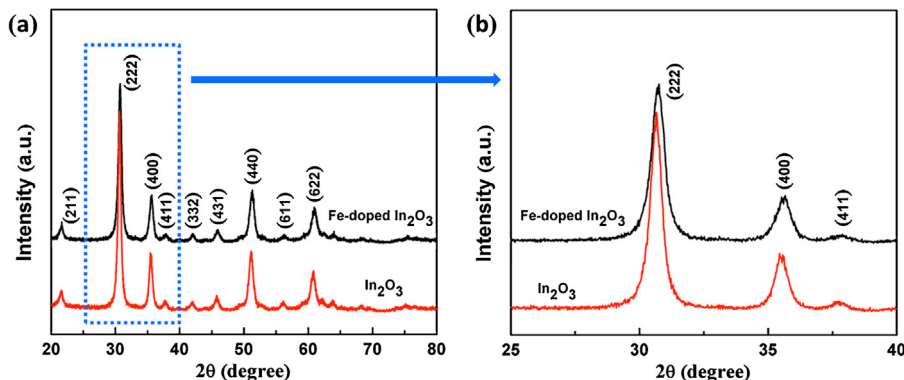


Fig. 3. (a) Wide-angle and (b) enlarged part XRD patterns of mesoporous  $\text{In}_2\text{O}_3$  and Fe-doped  $\text{In}_2\text{O}_3$ .

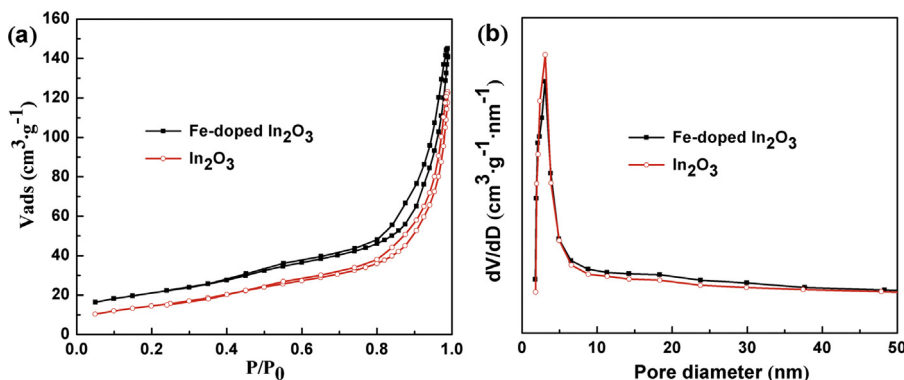


Fig. 4. (a) Nitrogen adsorption–desorption isotherms and (b) the corresponding pore size distribution of mesoporous  $\text{In}_2\text{O}_3$  and Fe-doped  $\text{In}_2\text{O}_3$ .

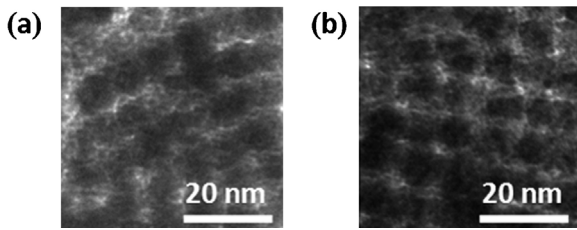


Fig. 5. TEM images of mesoporous (a)  $\text{In}_2\text{O}_3$  and (b) Fe-doped  $\text{In}_2\text{O}_3$ .

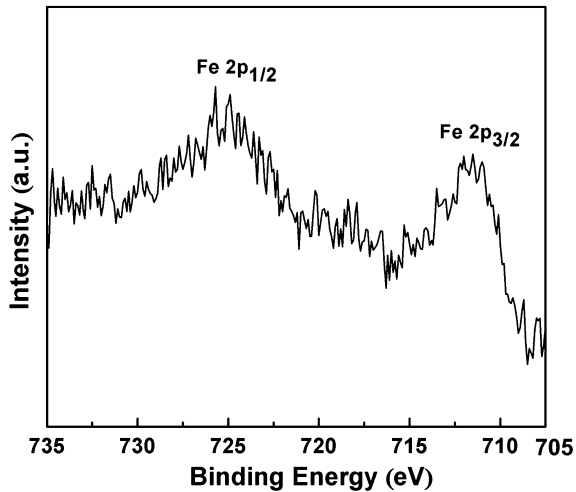


Fig. 6. XPS spectrum in the Fe 2p region of the mesoporous Fe-doped  $\text{In}_2\text{O}_3$ .

crystalline structure of the mesoporous  $\text{In}_2\text{O}_3$  and Fe-doped  $\text{In}_2\text{O}_3$  nanoparticles are obtained.

$\text{N}_2$  adsorption–desorption isotherms of mesoporous  $\text{In}_2\text{O}_3$  and Fe-doped  $\text{In}_2\text{O}_3$  samples are shown in Fig. 4(a). They all show typical type IV isotherms with hysteresis loops, which indicate that the obtained  $\text{In}_2\text{O}_3$  and Fe-doped  $\text{In}_2\text{O}_3$  samples exhibit the mesostructure. The corresponding pore size distribution of mesoporous  $\text{In}_2\text{O}_3$  and Fe-doped  $\text{In}_2\text{O}_3$  samples are displayed in Fig. 4(b). According to the BJH (Barrett–Joyner–Halenda) algorithm, the narrow pore size distribution of  $\text{In}_2\text{O}_3$  and Fe- $\text{In}_2\text{O}_3$  calculated from the adsorption branch has a mean value approximately 3.0 nm and 3.2 nm. The BET surface area and the total pore volume of the mesoporous Fe- $\text{In}_2\text{O}_3$  are calculated to be  $80.0 \text{ m}^2/\text{g}$  and  $0.24 \text{ cm}^3/\text{g}$ , respectively, which are somewhat higher than that of pure mesoporous  $\text{In}_2\text{O}_3$  ( $59.0 \text{ m}^2/\text{g}$  and  $0.22 \text{ cm}^3/\text{g}$ ).

The structures of mesoporous  $\text{In}_2\text{O}_3$  and Fe-doped  $\text{In}_2\text{O}_3$  were further characterized by TEM. In Fig. 5, the TEM images clearly

show that all the samples exhibit distinguishable periodic ordering mesostructure and highly crystalline nanoparticles, indicating the successful replication from the silica templates. This feature suggests that these particles of samples regularly assemble together inside the pores of the hard template. All these characterization results confirm that ordered crystalline mesoporous  $\text{In}_2\text{O}_3$  and Fe-doped  $\text{In}_2\text{O}_3$  are obtained with a nanocasting route using KIT-6 silica as a hard template. In particular, the mesoporous  $\text{In}_2\text{O}_3$  and Fe-doped  $\text{In}_2\text{O}_3$  replica after heating at  $500^\circ\text{C}$  for 3 h remains ordered mesostructure and high specific surface area, which suggests their good thermal stability.

The chemical state of the iron species has been investigated with X-ray photoelectron spectroscopy. As shown in Fig. 6, the binding energies of Fe  $2p_{3/2}$  and Fe  $2p_{1/2}$  are centered at 711.2 and 724.7 eV, respectively, indicating the existence of the  $\text{Fe}_2\text{O}_3$  [41,42].

### 3.2. Sensing properties

The response of the sensors based on mesoporous  $\text{In}_2\text{O}_3$  and Fe-doped  $\text{In}_2\text{O}_3$  samples toward 1 ppm  $\text{NO}_2$  were measured at various operating temperatures. As shown in Fig. 7(a), the maximum response of the sensor based on the mesoporous pure  $\text{In}_2\text{O}_3$  is 27 at  $100^\circ\text{C}$  and the response of mesoporous Fe- $\text{In}_2\text{O}_3$  sensor has reached to 116 at  $100^\circ\text{C}$ . Obviously, the sensor based on mesoporous Fe-doped  $\text{In}_2\text{O}_3$  exhibits much higher response than that of mesoporous pure  $\text{In}_2\text{O}_3$  at each temperature. It is well known that the response time is also important parameter for evaluating the properties of gas sensor. The correlation between the response and response time of the mesoporous Fe-doped  $\text{In}_2\text{O}_3$  sensor to 1 ppm  $\text{NO}_2$  with the operating temperatures are shown in Fig. 7(b). The response time decreased dramatically, with the increasing of operating temperature. Such behavior indicates that fast response can be achieved by increasing the operating temperature. Therefore, balance of the higher response and faster response speed of the mesoporous Fe-doped  $\text{In}_2\text{O}_3$  sensor, we have chosen  $150^\circ\text{C}$  as the optimum operating temperature, which is applied in all the investigations hereafter. The transient response of the mesoporous Fe-doped  $\text{In}_2\text{O}_3$  sensor to 1 ppm  $\text{NO}_2$  was measured at  $150^\circ\text{C}$  (Fig. 8), where the response is 71, as well as the response and recovery time is about 4.6 min and 2.5 min, respectively.

Fig. 9(a) shows the transient response of the mesoporous Fe-doped  $\text{In}_2\text{O}_3$  sensor sequentially exposed to 0.04, 0.1, 0.2, 0.5, 1, 2, 5 and 10 ppm  $\text{NO}_2$  at  $150^\circ\text{C}$ . The response of the sensor increased with the increasing of the  $\text{NO}_2$  concentration. It is worth noting that the sensor showed an obvious response to  $\text{NO}_2$  concentration even as low as 40 ppb, and the response has reached to 6. Correspondingly, the response of mesoporous Fe- $\text{In}_2\text{O}_3$  sensor as a function of  $\text{NO}_2$  concentration is displayed in Fig. 9(b). It increases rapidly with the increasing of gas concentration below 1 ppm. Though the

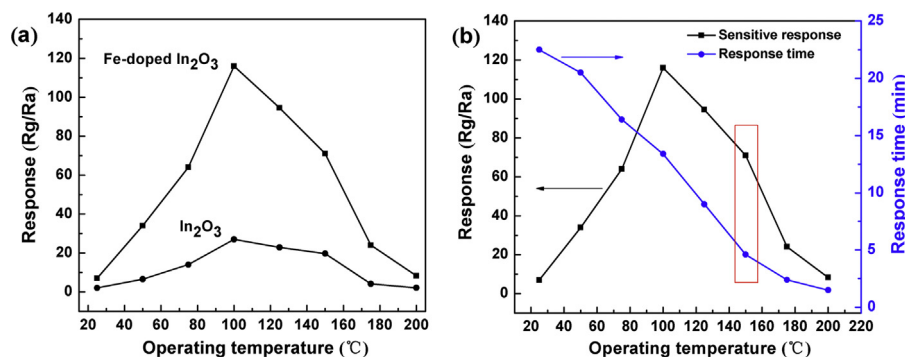


Fig. 7. (a) Response of the sensors based on mesoporous  $\text{In}_2\text{O}_3$  and Fe-doped  $\text{In}_2\text{O}_3$  samples toward 1 ppm  $\text{NO}_2$  at various operating temperatures and (b) Correlation between the response and response time of the mesoporous Fe-doped  $\text{In}_2\text{O}_3$  sensor to 1 ppm  $\text{NO}_2$  with the operating temperatures.

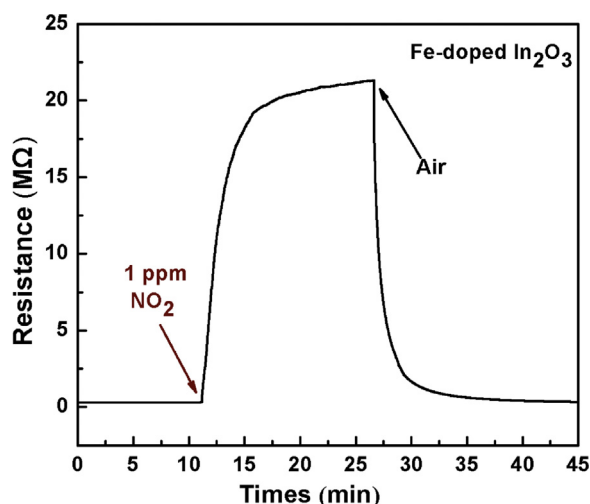


Fig. 8. Response and recovery of the mesoporous Fe-doped  $\text{In}_2\text{O}_3$  sensor to 1 ppm  $\text{NO}_2$  at  $150^\circ\text{C}$ .

response increases relatively slow when the  $\text{NO}_2$  concentration is more than 1 ppm, it is not saturated until the concentration is up to 10 ppm, which indicates that the sensor is suitable for detection of  $\text{NO}_2$  in a wide range. This unsaturation phenomenon to relatively high  $\text{NO}_2$  concentration may be resulted from the high surface area and large pore volume of mesoporous Fe-doped  $\text{In}_2\text{O}_3$  material, which provide abundant surface active sites and accommodate large number of target gas molecules.

The selectivity of the gas sensor is also important for practical application. Fig. 10 shows the cross-sensitive response of the mesoporous Fe-doped  $\text{In}_2\text{O}_3$  sensor to various gases, including  $\text{O}_3$ ,  $\text{Cl}_2$ ,  $\text{NH}_3$ ,  $\text{H}_2\text{S}$ ,  $\text{H}_2$ ,  $\text{CO}$  and  $\text{CH}_4$ . It is obvious that the mesoporous Fe- $\text{In}_2\text{O}_3$  sensor exhibits the largest response to  $\text{NO}_2$  among the tested gases. Such result indicates that the sensor using the mesoporous Fe- $\text{In}_2\text{O}_3$  exhibits an excellent selectivity to  $\text{NO}_2$  against the other tested gases.

The gas-sensing mechanism of n-type semiconductor sensor is mainly based on the electrical conductivity change originated from adsorption and desorption of gas molecules on the surface of sensing materials. When mesoporous Fe- $\text{In}_2\text{O}_3$  sensor is placed in air, the adsorbed oxygen molecules will capture electrons from the sensor to generate chemical adsorption oxygen species ( $\text{O}_2^-$ ,  $\text{O}^-$  and  $\text{O}^{2-}$ ). The oxygen vacancies of the sensor as intrinsic electron donors provide electrons to the conduction band and extrinsic acceptors created by chemisorbed oxygen species result in the formation of a thick space-charge layer and depletion of free electrons

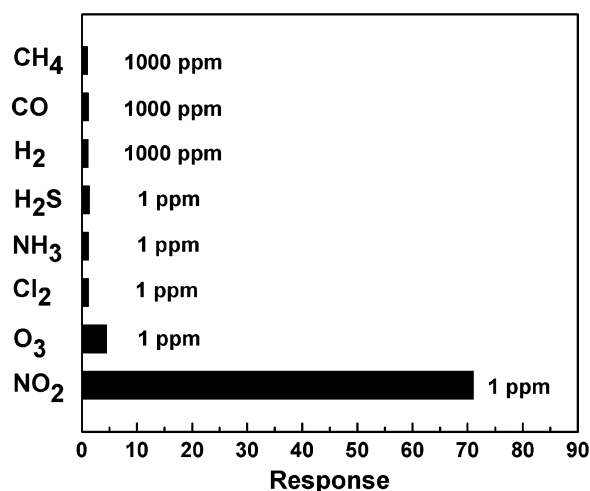
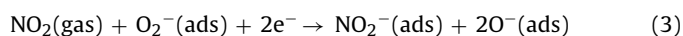


Fig. 10. Cross-sensitive response of the mesoporous Fe-doped  $\text{In}_2\text{O}_3$  sensor to various gases.

in the near-surface region of each grain (depletion layer), lowering the overall electrical conductivity (increasing the resistance). When mesoporous Fe-doped  $\text{In}_2\text{O}_3$  sensor is exposed to  $\text{NO}_2$  gas,  $\text{NO}_2$  can not only capture the electrons from the conduction band due to its higher electrophilic property but also react with the adsorbed oxygen species leading to the formation of adsorbed  $\text{NO}_2^-$  (ads). The mechanism can be explained by several chemical reactions which are shown below [2,43,44]:



The above reactions decrease the carrier concentration and electron mobility on the sensor surface, which lead to the increase of depletion layer width and accompany with the increase of resistance.

The enhanced sensing properties of the mesoporous Fe- $\text{In}_2\text{O}_3$  sensor may be caused by higher surface area which can provide more active sites to react with the target gas, whereas an ordered mesoporous structure and high pore volume facilitates the adsorption and diffusion of  $\text{NO}_2$  molecules. Moreover, incorporation of iron species on the sensor favors the effective adsorption of  $\text{NO}_2$  on the surface. All the factors are obviously beneficial for the enhancement in gas-sensing properties, and thus the mesoporous Fe-doped  $\text{In}_2\text{O}_3$  sensor exhibits a higher sensing response than that of mesoporous pure  $\text{In}_2\text{O}_3$  sensor.

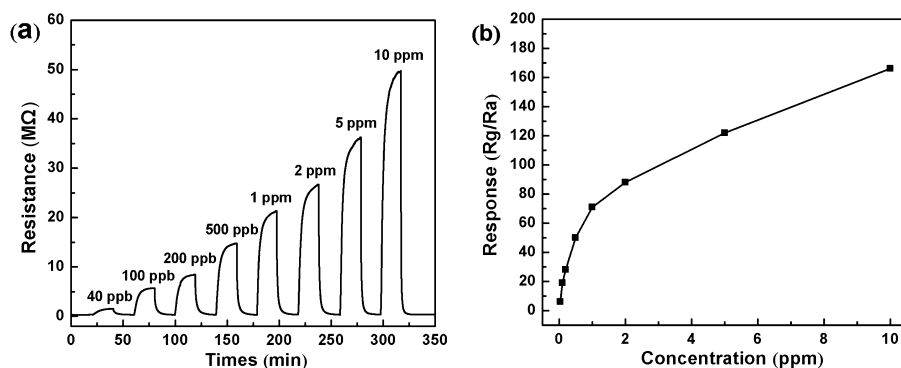


Fig. 9. (a) Transient response of the mesoporous Fe-doped  $\text{In}_2\text{O}_3$  sensor to different  $\text{NO}_2$  concentration at  $150^\circ\text{C}$  and (b) Response of mesoporous Fe- $\text{In}_2\text{O}_3$  sensor as a function of  $\text{NO}_2$  concentration.



#### 4. Conclusions

In summary, the utilization of mesoporous KIT-6 silica as a hard template has turned out to be a versatile tool for the synthesis of mesoporous Fe-doped  $\text{In}_2\text{O}_3$  in the nanocasting process. It exhibits high specific surface area, ordered periodically arranged mesoporous structure with well crystalline walls and high thermal stability. The product is a promising material for application as gas sensor, since its excellent physical and chemical properties. The gas sensing measurements reveal significantly increased sensitivity response to  $\text{NO}_2$  gas in comparison with the mesoporous pure  $\text{In}_2\text{O}_3$ . It is supposed that the outstanding gas sensing properties of the mesoporous Fe-doped  $\text{In}_2\text{O}_3$  sensor is attributed to higher specific surface area and the pore volume. Besides, Fe species loaded on the sensor is effective to adsorb  $\text{NO}_2$  gas on the surface, which is also beneficial to the sensing performance of the sensor.

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